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Electricity generation and bivalent copper reduction as a function of operation time and cathode electrode material in microbial fuel cells

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Abstract

The performance of carbon rod (CR), titanium sheet (TS), stainless steel woven mesh (SSM) and copper sheet (CS) cathode materials are investigated in microbial fuel cells (MFCs) for simultaneous electricity generation and Cu(II) reduction, in multiple batch cycle operations. After 12 cycles, the MFC with the CR cathode exhibits 55% reduction in the maximum power density and a 76% increase in Cu(II) removal. In contrast, the TS and SSM cathodes at cycle 12 show maximum power densities of 1.7 (TS) and 3.4 (SSM) times, and Cu(II) removal of 1.2 (TS) and 1.3 (SSM) times higher than those observed during the first cycle. Diffusional resistance in the TS and SSM cathodes is found to appreciably decrease over time due to the copper deposition. In contrast to CR, TS and SSM, the cathode made with CS is heavily corroded in the first cycle, exhibiting diminishment of both the maximum power density and Cu(II) removal at cycle 2, and being stabilized thereafter. These results demonstrate that the initial deposition of copper on the cathodes of MFCs is crucial for efficient and continuous Cu(II) reduction and electricity generation over prolonged time, which is closely associated with the nature of the cathode material. Among the cathode materials examined, the SSM is the most effective and inexpensive cathode candidate for practical use in MFCs.

Keywords: Microbial fuel cell; Cu(II) reduction; Electricity generation; Cathode electrode; Deposited copper

1 Introduction

Bioelectrochemical systems (BESs) including microbial fuel cells (MFCs) and microbial electrolysis cells (MECs) are regarded as new, sustainable and effective technologies for the recovery of metals from wastes and wastewaters [1]. Among the variety of heavy metals that can be recovered by BESs from wastes and wastewaters, the recovery of Cu(II) has attracted significant attention due to its wide presence in acid mine drainage wastewaters [2,3]. The reduction of Cu(II) can be achieved on the abiotic cathodes of two-chamber MFCs, since this metal has a relatively high redox potential (+0.286 V vs. standard hydrogen electrode, SHE) relative to the redox potential of the organic matter (ca. -0.30 V for acetate under standard conditions vs. SHE) in the anolyte [4]. MFCs with various operating volumes and experimental conditions, as well as different architectures, have been explored for more efficient Cu(II) reduction and higher electricity generation with varying degrees of success, as shown in Table 1 [3–13]. However, these studies do not investigate the performance of the cell in multiple batch cycles, and the long-term stability of MFCs for Cu(II) reduction, even though the longevity of the MFCs is crucial for its commercial application.

Table 1 Summary of the studies applying BESs for Cu(II) recovery, including the processing parameters used and the relevant recovery efficiencies obtained.

<table>
<thead>
<tr>
<th>Electron donor</th>
<th>Reactor</th>
<th>Anode</th>
<th>Cathode</th>
<th>Operation mode</th>
<th>Anodic working volume (L)</th>
<th>Cathodic working volume</th>
<th>Initial pH</th>
<th>Initial Cu(II) concentration</th>
<th>Operation time (h)</th>
<th>Reduction rate (mg L⁻¹ h⁻¹)</th>
<th>Product</th>
<th>Maximum power (W m⁻³)</th>
<th>Power produced (kWh kg⁻¹ Cu)</th>
<th>References</th>
</tr>
</thead>
</table>

This table summarizes the studies applying BESs for Cu(II) recovery, including the processing parameters used and the relevant recovery efficiencies obtained.
non-corrosive, carbon-based materials such as carbon cloth, carbon rod, graphite felt, graphite foil, graphite rod, graphite plate, as well as metals of titanium wire and copper plate have been used as cathodes in MFCs for Cu(II) reduction [3–13]. In view of a practical environmental application, during the prolonged operation of MFCs results in increasing amount of copper oxide unavoidable deposited on the surface of the cathodes, and as a result, the interaction of the cathode material with the deposited copper becomes crucial for determining the long-term performance of the cell. Copper is an excellent conductor that interacts with various materials. It promotes enhanced electrocatalytic activity towards oxygen and hydrogen peroxide reduction reactions in conventional electrochemical processes, due to the enlarged surface roughness and to the presence of new active sites. Furthermore, Films of CuO on stainless steel mesh also exhibit efficient catalytic activities for propene oxidation [17] while copper phthalocyanine performs as platinum catalysis for oxygen reduction reaction in MFCs [16]. Non-catalytic, carbon-based materials such as carbon cloth, carbon rod, graphite felt, graphite foil, graphite rod, graphite plate, as well as metals of titanium wire and copper plate have been used as cathodes in MFCs for Cu(II) reduction [3–13]. In view of a practical environmental application, during the prolonged operation of MFCs results in increasing amount of copper oxide unavoidable deposited on the surface of the cathodes, and as a result, the interaction of the cathode material with the deposited copper becomes crucial for determining the long-term performance of the cell. Copper is an excellent conductor that interacts with various materials. It promotes enhanced electrocatalytic activity towards oxygen and hydrogen peroxide reduction reactions in conventional electrochemical processes, due to the enlarged surface roughness and to the presence of new active sites [14–16]. Films of CuO on stainless steel mesh also exhibit efficient catalytic activities for propene oxidation [17] while copper phthalocyanine performs as platinum catalysis for oxygen reduction reaction in MFCs [16]. With regards to the capacitance of the electrodes, nanocrystals of copper can function with mesoporous activated carbons similarly as noble metals such as Pt and Pd, giving rise to substantial enhancement of the capacitance of carbonaceous electrodes [15]. Based on the intrinsic excellent characteristics of copper, Motos et al. [10] successfully decreased the internal resistance of MFCs through simply coupling a copper plate cathode together with an anion exchange membrane and a carbon felt anode, and thus achieve a copper reduction rate of 75 mg L⁻¹ h⁻¹ and a maximum power production of 5.5 W m⁻³ at a high initial Cu(II) concentration of 2000 mg L⁻¹ [10] (Table 1). The metal reduction rates of Cu(II) reduction and power production would be reasonably expected to increase in proportion to the concentration of Cu(II) in the water. Together with considering that the performance of the electrode materials commonly used in MFCs including carbon rod (CR), titanium sheet (TS) and stainless steel woven mesh (SSM), their significant influence on the deposited copper, with these excellent characters may affect which its excellent physico-chemical characteristics may alter the electrochemical behaviour of the surface of the cathode and thus alter the and, as a result, the overall subsequent performance of MFCs which to our knowledge This effect has not been systematically investigated in literature. In this study, species specimens of CR, TS and SSM are investigated as cathode materials in MFCs for simultaneous electricity generation and Cu(II) reduction, in multiple batch cycle operations. For comparison convenience.
copper sheet (CS) is additionally used as one further cathode material to assess the MFC system performance. The effect of the deposited copper on system performance is elucidated by linear sweep voltammetry (LSV), scanning electronic microscopy (SEM), energy dispersive X-ray spectrometry (EDS), X-ray diffraction (XRD) and electrochemical impedance spectroscopy (EIS). The maximum power density and Cu(II) reduction in the MFCs were compared with those obtained with different batch cycle operations under otherwise identical conditions. The circuital current, the anodic columbic efficiency (CE) and the cathodic CE were employed to assess system performance.

2 Materials and methods

2.1 Reactor setup

Two-chamber MFCs (duplicates) were used in all experiments, with the chambers separated by a cation exchange membrane (CMI-7000 Membranes International, Glen Rock, NJ). The anodes were made with porous graphite felt (1.5 x 1.5 x 1.0 cm, Sanye Co., Beijing, China). The working volumes of the anode and cathode compartments were 20 mL each. The cathodes were made with CR (Chiji Duraflight Carbon Co., China), TS, SSM and CS (Qingyuan Co., China) and were all exposing the same surface area of 8 cm². The compositions of cathode electrodes were listed in Table S1. The cathode materials were first mechanically polished with abrasive papers and then cleaned with ethanol and deionized water before their installation in the MFCs.

2.2 Inoculation and operation

The anodes were inoculated from the anodes of previous operating MFCs running on acetate for faster anodic biofilm acclimation [3,11,19,20]. The composite of anolyte was as previously described [21]. The anolyte was sparged with ultrapure N₂ gas for 15 min, prior to transfer into the reactors. For anode acclimation, deionized water was used as catholyte and CR as cathode electrodes, with an external resistor of 510 Ω. After six-cycle anolyte refreshments with each lasting 2–3 days, the catholyte was replaced by aqueous CuCl₂ at a Cu(II) concentration of 50 mg L⁻¹. Prior to adding the Cu(II)-catholyte into the cathode chambers, the catholyte was thoroughly sparged with ultrapure N₂ gas for 15 min in order to exclude the effect of dissolved oxygen on Cu(II) reduction [4].

The acclimation period was completed after another 2–3 refreshments with stable and repeatable voltage output. Cathodic LSV was performed from cathodic open electrode potential (0.45 V) to 0.0 V at a scan rate of 0.1 mV s⁻¹. Cathodic LSV and EIS were performed using the same potentiostat with three electrode system consisted of a working electrode (cathode electrode) in the cathode, an Ag/AgCl reference electrode (195 mV vs. SHE) and a counter electrode (cathode electrode) in the cathode, an Ag/AgCl reference electrode (195 mV vs. SHE) setting 1 cm away from the cathode in the cathodic chamber and a Pt foil (2 x 4 cm) counter electrode in the anodic chamber. The cathodic LSV was performed from cathodic open electrode potential (0.45 V for CR, 0.22 V for TS and 0.32 V for SSM) to −0.08 V (vs. SHE) at a scan rate of 0.1 mV s⁻¹. Since working potentials of all cathode electrodes were more positive than −0.08 V (vs. SHE), the lowest scanned potential of −0.08 V (vs. SHE) can guarantee the range of all cathodic potentials under the experimental conditions.

The catholyte was always composed of 50 mg L⁻¹ Cu(II) at an initial pH of 2.0 and a solution conductivity of 5.0 mS cm⁻¹. This low initial pH of 2.0 was more approachable to practical acid mine wastewaters [2,3]. All the potentials reported in this study were in volts (V) units relative to a SHE. One-way ANOVA in SPSS 19.0 was used to analyze the differences among the data, and all of the data indicated significance levels of p < 0.05.
where $I$ is the circuit current (A); $t$ is a set operational time (s); $\Delta \text{COD}$ is the corresponding cumulative COD values in the anolytes over the operational period of $t$ hours (g L$^{-1}$); $V_{an}$ and $V_{ca}$ are the anode and cathode working volumes, respectively (L); $\Delta C$ is the change of Cu(II) concentrations during the operational time $t$ (mg L$^{-1}$); 2 and 4 are the molar numbers of electrons required for Cu(II) reduction and oxygen reduction, respectively (mol mol$^{-1}$); 65 is the molecular weight of copper (g mol$^{-1}$); 32 is the molecular weight of O$_2$ (g mol$^{-1}$) and $F$ is the Faraday constant (96485 C mol$^{-1}$ e$^-$).

### 3 Results and discussion

#### 3.1 Polarization curves as a function of time

Maximum power density with the CR cathode gradually increased from 1.9 W m$^{-3}$ (7.1 A m$^{-3}$) at cycle 2 to 2.6 W m$^{-3}$ (9.4 A m$^{-3}$) at cycle 12 (Fig. 1A). An exceptional high power of 3.7–4.6 W m$^{-3}$ (7.6–24.3 A m$^{-3}$) was observed at cycle 1, mainly ascribed to the occurrence of a power overshoot probably due to ions and substrate limitations [24]. Similarly, TS (Fig. 1D) and SSM (Fig. 1G) cathodes exhibited a gradual increase in the maximum power densities, reaching 3.1 W m$^{-3}$ (11.6 A m$^{-3}$) and 6.5 W m$^{-3}$ (26.9 A m$^{-3}$) in cycle 12, 1.7 and 3.4 times as high as those in cycle 1. These results illustrate the dependency of the maximum power density on both cathode material and operational time. At cycle 14, no further increase in the maximum power density was observed for all CR, TS and SSM cathodes (Fig. 1A, D, G and J). While an insoluble copper crust had sufficiently covered the cathodes at cycle 12, the reduction of some deposited copper from the SSM cathodes at cycle 14 explained the increase in ohmic resistance (Fig. 1G and H). In contrast to these results, CS cathode showed a significantly higher maximum power of 10.8 W m$^{-3}$ (33.8 A m$^{-3}$) at cycle 1 (Fig. 1J) which decreased to and stabilized at 3.2 W m$^{-3}$ (10.6 A m$^{-3}$) over successive cycles, implying the occurrence of heavy CS corrosion at cycle 1. This result was consistent with the very recent report of Cu(II) reduction on the copper plate cathodes in MFCs [10].
Concomitant with power density, the OCPs with CR cathodes sharply decreased from 0.65 V at cycle 1 to 0.40 V at cycle 2, stabilizing at 0.40–0.43 V thereafter (Fig. 1B), whereas cathodes with TS (Fig. 1E) and CS (Fig. 1K) always exhibited OCPs in the range 0.40–0.43 V over time. Although SSM cathodes produced higher OCPs of 0.50–0.52 V during the initial 8 cycles, the overpotentials during this period was apparently high (Fig. 1H), resulting in a comparatively lower power production (Fig. 1G).

The cathodic potentials of all cathode materials varied much more than the anodic potentials over the current density range (Fig. 1C, F, I and L), implying that the cathode rather than anode was controlling the performance of these two-chamber MFCs as a function of time. These results give clear evidence that the association of the deposited copper with the cathode surfaces alters the propensity of these materials for electricity generation and voltage output during prolonged operation

3.2 Circuital current, Cu(II) removal and columbic efficiency (CE) over time

In general, both SSM and TS cathodes exhibited gradual increases in the circuital current (Fig. 2A), Cu(II) removal (Fig. 2B) and anodic CEs (Fig. 2C) over time. The Cu(II) removal at cycle 12 was 99.7 ± 0.4% (SSM) and 87.7 ± 1.5% (TS), which were as high as 1.28 (SSM) and 1.20 (TS) times of those in cycle 1. These results are in agreement with the results of the polarization tests (Fig. 1D, E, G and H). CR cathodes in cycle 1 had a low Cu(II) removal of 48.8 ± 1.9% (Fig. 2B) compared to the highest circuital current of 0.230 ± 0.001 mA (Fig. 2A), thus leading to low cathodic CEs (Fig. 2D). CS cathodes always showed the highest cathodic CEs than the others (Fig. 2D), ascribed to the lowest circuital current (Fig. 2A) and higher Cu(II) removal (Fig. 2B). Except CR, the observation of cathodic CEs with other material cathodes in excess of 100% at cycle 1 (Fig. 2D) indicates the occurrence of cathode corrosion, which was more significant in CS cathodes. The occurrence of cathode corrosion was mainly ascribed to the presence of Cl⁻ in the catholyte [25]. Other researchers also observed the corrosion of the cathodes or anodes in MFCs [26–28]. The decreasing cathodic CEs (Fig. 2D) and increasing Cu(II) reduction (Fig. 2B) for TS and SSM over consecutive cycles reflects the rates of corrosion decline and an increase in direct electron transfer. Thus, for TS and SSM, there was a shift away from corrosion to a predominantly cathodic Cu(II) reduction over time, although it was not possible to discern the relative contribution that these two different processes may have had on Cu(II) reduction. No Cu(II) was measured in the anolyte during these tests, although its existence in the cation exchange membrane cannot be excluded [11].

Fig. 1 Power density (A, D, G and J), voltage output (B, E, H and K), and anodic and cathodic potentials (C, F, I and L) for MFCs with cathodes of CR (A, B and C), TS (D, E and F), SSM (G, H and I), and CS (J, K and L) as a function of time (each cycle lasted 6 h).
The relative contribution that these two different processes may have had on Cu(II) reduction.

Cu(II) was not detected in the anolyte during these tests, although its existence in the cation exchange membrane could not be excluded.

The main advantage of using SSM cathodes in comparison to TS is the lower cost. The purchasing cost of the SSM used here was around 73 $ \text{m}^{-2}$, compared to 450 $ \text{m}^{-2}$ for TS. In view of flexibility, SSM is certainly more suitable than CR for different MFCs configurations, producing a higher circuital current and a more efficient rate of Cu(II) reduction. Considering all the factors, this study shows that the SSM cathode has more advantages over the other cathode materials for application in larger-scale MFCs systems, showing the most appreciable rate of Cu(II) reduction, more flexibility than CR for different MFCs configurations, lower cost than TS, and much less corrosion than CS.

3.3 Electrochemical characterization using EIS

EIS spectra were fitted to equivalent circuits (Fig. S1) to identify the components of the internal resistances of the various cathode material cathodes during cycles 1 and 12 (Fig. 3). In cycle 1, TS had exhibited a higher electrode resistance (629 $\Omega \text{cm}^2$) than SSM (579 $\Omega \text{cm}^2$), both of which were much higher than CR (169 $\Omega \text{cm}^2$) and CS (143 $\Omega \text{cm}^2$) (Fig. 4). In cycle 12, however, these values substantially decreased to 240 $\Omega \text{cm}^2$ (TS) and 223 $\Omega \text{cm}^2$ (SSM), both of which were similar to 223 $\Omega \text{cm}^2$ for CR at the same cycle, explaining the improved system performances of the TS and SSM cathodes over time (Figs. 1 and 2). In contrast, CR had a slight increase in resistance from an initial 169 $\Omega \text{cm}^2$ to 223 $\Omega \text{cm}^2$ at cycle 12, consistent with the low values in both circuital current (Fig. 2A) and Cu(II) removal (Fig. 2B). These results are supported by the observation of Zhang et al. [15], where a copper content of 12% increased the resistance of the substratum of activated carbon. Resistances of CS cathodes were always similarly low, mainly ascribed to the copper intrinsic characteristics of excellent conductor [10].
The appreciable decrease in the diffusion resistance ($R_d$) was observed with the SSM and TS cathodes at cycle 12 (Fig. 4), considerably contributed to the apparent decrease in internal resistance, consistent with previous reports of enhanced surface conductivity by deposited copper in conventional electrochemical processes [14–16]. Similarly, the polarization resistance ($R_p$) and ohmic resistance ($R_o$) with both SSM and TS cathodes also diminished with copper deposition over time. These results illustrate the favorable effect of the deposited copper on lessening the internal resistance of the SSM and TS electrodes.

3.4 Electrochemical analysis of cathodes using LSV

A strong reduction peak at a potential of 0.08 V and a current of −5.95 mA was observed for TS cathode, followed by 0.08 V and −3.49 mA for SSM at the same cycle 12 (Fig. 5). These results support the finding of the circuital current (Fig. 2A) and Cu(II) removal (Fig. 2B), where TS reached the highest, followed by SSM, and CR had the lowest. Much lower reduction peaks observed with the CR than TS and SSM at the same cycle 12 implies less incorporative and smaller interactional effects of the copper deposited on CR on the exchange current. Control experiments in the absence of deposited copper reported reduction peaks with much lower currents, reflecting the tremendous importance of the deposited copper in developing high current peaks, consistent with the excellent conducting properties of copper.
3.5 Morphology of the cathode and product confirmation

Compared to the bare cathodes of CR (Fig. 6A), TS (Fig. 6D), SSM (Fig. 6G) and CS (Fig. 6J), a layer of typical reddish-brown color of pure copper was observed on the electrodes after the completion of cycle 1 (Fig. 6B, E, H and K), implying the successful reduction of Cu(II) to pure copper on the surface of the various electrodes [3,5,9,19,20]. In contrast, the cathodes exhibited a darker appearance after exposure to air for 48 h (Fig. 6C, F, I and L), which resulted from the oxidation of metallic copper and the consequent alteration of both physical and electrochemical properties of the electrodes.
The concentration of copper ion in solution with CS cathodes was further evaluated under open circuit conditions and in the absence of Cu(II) in the catholyte. The concentration of Cu(II) in the catholyte at the end of the experiment was found to be 82 ± 1 mg L⁻¹ with simultaneous reduction in the CS net weight by 22 ± 1 mg, confirming the occurrence of CS corrosion as the experiment proceeded. In parallel, the color of the catholyte solution changed from colorless to light blue (Fig. S2). These results are in agreement with previous studies with MFCs using copper anodes or cathodes [10,27]. In contrast, no color change was observed in the fresh catholyte under closed circuit conditions, illustrating the protecting role of cathodic electrons on the surface of the sacrificial copper cathode in this system.

Diverse nano-metric scale agglomerates were found on all cathodes of CR (Fig. 7A), TS (Fig. 7D), SSM (Fig. 7G) and CS (Fig. 7J) at the end of cycle 1, which provided a higher interfacial surface area for Cu(II) adsorption and an increase of the conductivity throughout the cathode [29]. These results were in good agreement with the improved electricity generation (Fig. 1A, D and G) and circuitual current (Fig. 2A) in the subsequent cycles. At the end of cycle 1, the distribution of copper nanoparticles on the surface of the CR cathode (Fig. 7A) showed a more uniform distribution than those on TS (Fig. 7D) and SSM (Fig. 7G) cathodes, mainly due to the high surface area of the carbon structure [18]. Rose-like (CR) (Fig. 7B), multi crystal (TS) (Fig. 7E) and cauliflower-like (SSM) (Fig. 7H) segregates were clearly observed after cycle 12, illustrating the importance of the cathode material on the diversity of shapes and morphology of the deposited copper [30–32]. The marked enhancement of the surface roughness after cycle 12 (Fig. 7B, E and H for CR, TS and SSM, respectively) in comparison to the smoother and compact surfaces after cycle 1 (Fig. 7A, D and G), led to an increase in the active surface area and, consequently, in the observed power production density (Fig. 1) and Cu(II) reduction (Fig. 2) [14]. In contrast, the CS cathode was severely corroded, resulting in remarkable sculptures on the surface, even after cycle 1 (Fig. 7J). More severe corrosion was reasonably observed on the CS cathode after cycle 12 (Fig. 7K). This result was similar to the corrosion of copper anodes, and MnO₂ and copper cathodes in MFCs, as well as magnetite cathodes in MECs [26–28]. Although the size and shape of the copper particles deposited on these various material cathodes materials were not tightly controlled, improvements in system performance were nonetheless observed, stressing the importance of the deposited copper on the cathode, particularly for subsequent cycles of Cu(II) reduction and electricity generation. A much deeper understanding of the issues affecting the morphology and nucleation of copper elements on the surfaces of each cathode materials is encouraged in further studies [31,32].
Further EDS analysis of the composition of the agglomerates, reported Cu signals at the same binding energies of 0.98, 8.06 and 8.87 keVs on the CR (Fig. 7C), TS (Fig. 7F), SSM (Fig. 7I) and CS (Fig. 7L) cathodes, confirming the formation of Cu product. The observation of Cl signals on all the cathodes was associated with the catholyte of CuCl₂. Stronger Cl signals appeared on the CS after a prolonged operational time at cycle 12 (Fig. S3), confirming again the occurrence of a more severe corrosion of the surface. While copper was deposited on the cathodes of CR, TS and SSM cathodes, it may be necessary to use a settling clarifier/bedding to remove and peel off the copper precipitates from the electrodes in order to achieve continuous copper removal in a practical system. Further investigations in this direction are certainly warranted.

The XRD patterns recorded on the CR, TS and SSM cathodes similarly and closely matched that of metal Cu₃ with standard peaks at 111, 200 and 220° in 2θ (Fig. 8A), demonstrating the invariable products and their independence on these cathode materials.

**Fig. 7** SEM micrographs of the cathodes of CR (A and B), TS (D and E), SSM (G and H), and CS (J and K) at the end of cycle 1 (A, D, G and J) and cycle 12 (B, E, H and K). EDS analysis on products of Cu(II) reduction on the cathodes of CR (C), TS (F), SSM (I) and CS (L) at the end of cycle 1.
It is generally agreed that the copper products formed on the cathodes are pH dependent. Under the experimental temperature of 20 ± 3 °C and at an initial Cu(II) concentration of 50 mg L⁻¹, the theoretical cathode potentials of Cu(II) reduction at different pHs were calculated based on the Nernst equations (Fig. 8B). Considering the final pHs of 2.2–3.4 observed in the MFCs, the Cu(II) species in MFCs was reasonably reductively changed to pure copper under the present experimental conditions, further confirming the final products in this system (Fig. 8A).

Since Cu(II) laden wastewater generally comprise a range of multiple metal ions such as Cr(VI) and Cd(II) and others, future studies should focus on the impact of these other metals on the MFCs performance. The positive effects of the deposited copper on subsequent Cu(II) reduction and electricity generation, in the MFCs fitted with the CR, TS and SSM cathodes should be investigated in the presence of non-conductive metal deposits such as those from chromium and cadmium [22,33], to determine the most efficient cathode materials in industrial scale MFCs.

4 Conclusions

Copper recovery from aqueous Cu(II) is more attractive than treating and recovering other heavy metals due to its wide occurrence in acid mine drainage wastewaters. Previous studies have primarily examined the performance of carbon-based or copper plate cathode for Cu(II) reduction in MFCs. In this study we have examined a range of cathode materials on the effectiveness of Cu(II) reduction and electricity generation in MFCs in multiple operational cycles. The results shows the TS and SSM cathodes as the most effective for Cu(II) removal and power generation with the performance increasing over multiple cycles. This effect was associated to a reduction of the ohmic resistance over the cathodes and the increase in interfacial surface area of the cathode. This study gives a comprehensive appreciation of the effect of deposited copper on the various material cathodes, which further has a profound effect on the efficiency of Cu(II) reduction and electricity generation over time. Overall, the SSM was found the most promising cathode material considering the observed high performance and low material cost for use in practical MFCs. However, further studied should shed more light on the effect of concomitant heavy metal and their deposits on the overall performance of MFCs for water remediation and simultaneous production of renewable energy.

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Appendix A. Supplementary data

Supplementary data related to this article can be found at http://dx.doi.org/10.1016/j.jpowsour.2016.01.022.

References


[31]
Appendix A. Supplementary data

The following is the supplementary data related to this article:

Multimedia Component 1

Graphical abstract

Highlights

• TS and SSM cathodes showed increased power production and Cu(II) removal over time.

• CR exhibited decreased power production and increased Cu(II) removal over time.

• TS and SSM cathodes exhibited a reduction of ohmic resistance over time.

Queries and Answers

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